

Directional Emission from Leaky and Guided Modes in GaAs Nanowires Measured by Cathodoluminescence

Benjamin J. M. Brenny,[†] Diego R. Abujetas,[‡] Dick van Dam,[¶] José A. Sánchez-Gil,[‡] J. Gómez Rivas,^{¶,§} and Albert Polman^{*,†}

[†]*Center for Nanophotonics, FOM Institute AMOLF, Science Park 104,
1098 XG Amsterdam, The Netherlands*

[‡]*Instituto de Estructura de la Materia (IEM-CSIC), Consejo Superior de Investigaciones Científicas, Serrano 121, 28006, Madrid, Spain*

[¶]*COBRA Research Institute, Eindhoven University of Technology, P.O. Box 513, 5600 MB Eindhoven, The Netherlands*

[§]*FOM Institute DIFFER, P.O. Box 6336, 5600 HH Eindhoven, The Netherlands*

E-mail: polman@amolf.nl

Abstract

1
2 We measure the polarization-resolved angular emission distribution from thin and
3 thick GaAs nanowires (diameters ~ 110 and ~ 180 nm) with cathodoluminescence po-
4 larimetry. The nanowires, which horizontally rest on a thin carbon film, are excited
5 by a 5 keV electron beam and emit band gap luminescence at a central wavelength of
6 870 nm. The emission can couple to different waveguide modes that propagate along
7 the wire, are dependent on the wire diameter and determine the directionality and
8 polarization of the emission. Although each measured nanowire can support different

9 modes, the polarized emission is dominated by the TM01 waveguide mode in all cases,
10 independently of wire diameter. When exciting the nanowires close to the end facets,
11 the thin and thick wires exhibit opposite directional emission. The emission from thin
12 nanowires is dominated by a leaky TM01 mode that leads to emission towards the
13 opposite end facet (emission to the right when exciting the left-side edge). For the
14 thick wires, however, the TM01 mode is guided but also lossy due to absorption in
15 the substrate. In such a case, the wires emit towards the excited end facet (to the left
16 when exciting the left-side edge). The emission directionality switches for nanowire
17 diameters in the range of 145-170 nm. We show that the measurements agree well
18 with both a simple 1D current model and numerical simulations. The high spatial res-
19 olution of angle- and polarization-resolved cathodoluminescence spectroscopy provides
20 detailed insight into the nanoscale emission and propagation of light in semiconductor
21 nanowires.

22 **Keywords**

23 Cathodoluminescence, nanowires, GaAs, polarimetry, waveguide modes

24 Semiconductor nanowires have fueled a growing field of integrated nanoscale optoelec-
25 tronic devices, such as lasers,¹⁻³ light emitting diodes,^{4,5} photovoltaics,⁶⁻⁹ single-photon de-
26 tectors,¹⁰⁻¹³ photodetectors¹⁴ and metamaterials.^{15,16} Both the electrical and optical prop-
27 erties of nanowires are eminently tunable by controlling their size, geometry or composition,
28 among others.¹⁷⁻²⁰ The directionality and polarization of emitted radiation from nanowires
29 have been examined in previous studies^{14,21-23} and result from the coupling to leaky and
30 guided waveguide modes,²⁴⁻²⁹ which can also be described by Mie and Fabry-Pérot reso-
31 nances.^{29,30} All modes are highly dependent on nanowire diameter.

32 Most previous studies of semiconductor nanowire emission properties have employed opti-
33 cal excitation methods. While powerful, such techniques lack the nanoscale spatial resolution
34 to uncover all the features of the radiative processes from these nanostructures. Here we

35 use cathodoluminescence (CL) spectroscopy, in which an electron beam acts as a highly lo-
36 calized excitation source and the emitted light is detected.^{31–33} The high spatial excitation
37 resolution of CL is typically determined by the electron beam spot size and the evanescent
38 field extent about the beam path ($\sim 10\text{--}30$ nm),³³ which enables the study of the nanoscale
39 modal behavior of light.^{34–37} In general, CL also allows the characterization of a wide range of
40 material properties.^{38–40} Recently, the ability to measure both the angular and polarization
41 distribution in CL has been demonstrated.^{41,42}

42 In this article, we use these new CL features to investigate the angle- and polarization-
43 dependent emission from horizontal GaAs nanowires.⁴¹ We study nanowires of different
44 lengths and diameters that support both leaky and guided modes. Exciting the nanowires
45 along their length, we find that the TM₀₁ mode dominates the polarization-resolved emission
46 for all excited wires, but depending on the diameter, the mode is either leaky or guided. We
47 observe a distinct change of the directionality of the CL emission when exciting the nanowires
48 close to their end facets, which correlates with the nanowire diameter and the nature of the
49 mode. Thinner, leaky wires emit in the opposite direction from thicker, guided wires. The
50 measurements exhibit good agreement with both a simple 1D current model and numerical
51 simulations, which show that the substrate also plays a role in the emission directionality.

52 Experiment

53 GaAs nanowires were grown by self-catalyzed molecular beam epitaxy on silicon^{43,44} and
54 were subsequently mechanically broken and deposited on a holey carbon TEM grid (see
55 Methods). Scanning electron micrographs of the two GaAs nanowires studied here are shown
56 in Figure 1(a). The thin NW1 has a length of $7.9\ \mu\text{m}$ and a diameter of $100\text{--}120$ nm, while
57 the thick NW2 has a length of $12\ \mu\text{m}$ and a diameter of $175\text{--}195$ nm. Both wires are slightly
58 tapered, the right-hand side being thinner, although NW2 does thicken again slightly at
59 the very edge. The nanowires lie horizontally on the ~ 20 nm thick carbon layer. In the

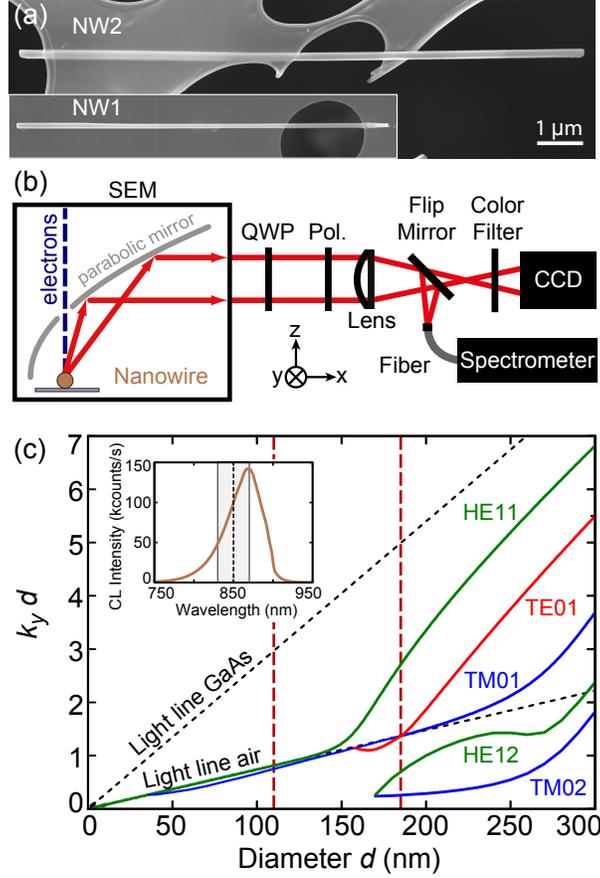


Figure 1: (a) Scanning electron micrographs of the GaAs nanowires NW1 (bottom) and NW2 (top), shown on the same scale. NW1 is 7.9 μm long and 100–120 nm thick; NW2 is 12 μm long and 175–195 nm thick. (b) Schematic overview of the cathodoluminescence polarimetry setup. The electron beam excites the nanowires, the emitted radiation is collected by a parabolic mirror and either focused onto a fiber connected to a spectrometer or sent through a QWP, linear polarizer and bandpass filter before being imaged onto a 2D CCD camera. (c) Dispersion relation of leaky and guided modes for infinitely long cylinders, showing the real part of the wavevector k_y multiplied by the cylinder diameter d , as a function of d , for GaAs at $\lambda_0 = 850$ nm ($n = 3.6$, $k_0 = 7.39 \mu\text{m}^{-1}$). The vertical red dashed lines indicate the average diameters of the two wires. The inset shows the measured CL emission spectrum from NW2. The spectrum of NW1 (not shown) does not differ noticeably except for a lower intensity. The vertical black dashed line in the inset at $\lambda_0 = 850$ nm indicates the transmittance maximum of the bandpass filter used for the angular measurements, while the gray area indicates the 40 nm bandwidth of the filter.

60 Supporting Information we show data for an additional thin and thick wire (SEM images
 61 shown in Figure S1).

62 The cathodoluminescence spectroscopy and polarimetry setup^{41,45,46} is schematically

63 shown in Figure 1(b). A parabolic mirror collects the radiation from the nanowires and
 64 directs it onto a spectrometer or images it onto a 2D camera to measure the angular in-
 65 tensity distribution for a given wavelength (using a bandpass filter). Polarization-resolved
 66 measurements are obtained by using a polarimeter composed of a quarter-wave plate (QWP)
 67 and a linear polarizer (Pol.), which determines the Stokes parameters of the emitted radi-
 68 ation. The full polarization can be detected in this way, obtaining information about the
 69 degree of polarization, its orientation, ellipticity, and handedness. Essentially, this allows the
 70 retrieval of any arbitrary polarization state, including the different electric field components
 71 and the phase difference between them.⁴⁷ This is not possible using only a linear polarizer.
 72 We correct for the geometrical and polarization dependent transformations of the parabolic
 73 mirror on the measured emission⁴¹ (see Methods for more details about the CL measure-
 74 ments). For the measurements, the nanowires are aligned along the y-axis, as defined by
 75 the coordinate system shown in Figure 1(b). As we expect directional emission along the
 76 nanowire axis, this is the preferred orientation for the mirror to collect the radiation sym-
 77 metrically. The CL emission spectrum from NW2 is shown in the inset of Figure 1(c) and
 78 is dominated by band-gap recombination centered around $\lambda_0 = 870$ nm. This emission can
 79 feed into waveguide modes supported by the nanowire, that depend on its diameter, and
 80 which can affect the polarization and directionality of the emitted radiation.^{22-24,28}

81 **Nanowire waveguide modes**

82 Figure 1(c) shows the dispersion relation of waveguide modes for infinitely long cylinders,⁴⁸
 83 calculated for GaAs at $\lambda_0 = 850$ nm ($n = 3.6$, $k_0 = 7.39 \mu\text{m}^{-1}$), the wavelength at which
 84 we filter the angle-resolved measurements. We follow the formalism used in Ref. 27 and
 85 determine the wavevector k_y along the axis of the nanowire. We show the real part of k_y
 86 multiplied by the wire diameter d , as a function of d . The dispersion curves denote transverse
 87 electric (TE), transverse magnetic (TM), and magnetoelectric (HE) modes. These modes are

88 characterized as “leaky” if their dispersion lies below the light line of air ($k_y < k_0$), in which
89 case they also possess a non-negligible imaginary part of the wavevector k_y .²⁷ If the mode
90 dispersion lies above the light line of air but below the light line in GaAs ($k_{GaAs} > k_y > k_0$),
91 the waveguide mode is guided within the nanowire.

92 The vertical red dashed lines in Figure 1(c) display the average diameter of the two
93 nanowires studied here. For the thin NW1 (and any nanowire with a diameter below 150
94 nm), only the TM01 and the HE11 modes are supported. Both are very close to the light
95 line in air, but the TM01 mode is slightly below it and thus leaky for these diameters. In
96 the case of the thicker NW2, the TE01, HE12 and TM02 also occur. The latter two are far
97 below the light line in air and thus have a very short propagation length along the nanowire,
98 while the HE11 mode is very clearly guided. The TE01 and TM01 modes are both very
99 close to the light line of air in this region, representing a transition region between a leaky
100 and guided nature for these modes. Which modes will dominate the emission depends on
101 the coupling efficiency between the excitation source and the mode.

102 The dispersion relation allows us to determine which modes can play a role in the emission
103 from these nanowires and to calculate the wavevector corresponding to each mode for a
104 given diameter. A 1D current model, developed in Ref. 27 and applied in Ref. 23, uses the
105 wavevectors to calculate far field emission patterns for all electromagnetic field components.
106 The model describes the nanowire as a 1D cavity in vacuum with length L ; the emission is
107 produced by a line current excited by a dipole at a given position along the wire. This simple
108 model allows us to retrieve the expected polarization-dependent angular emission patterns
109 for different modes at different wire diameters, which we can compare to measurements.

110 **CL polarimetry**

111 Measurements and calculations of the angle- and polarization-dependent emission intensity
112 distributions at $\lambda_0 = 850$ nm for central excitation of the two nanowires and clearly recognize

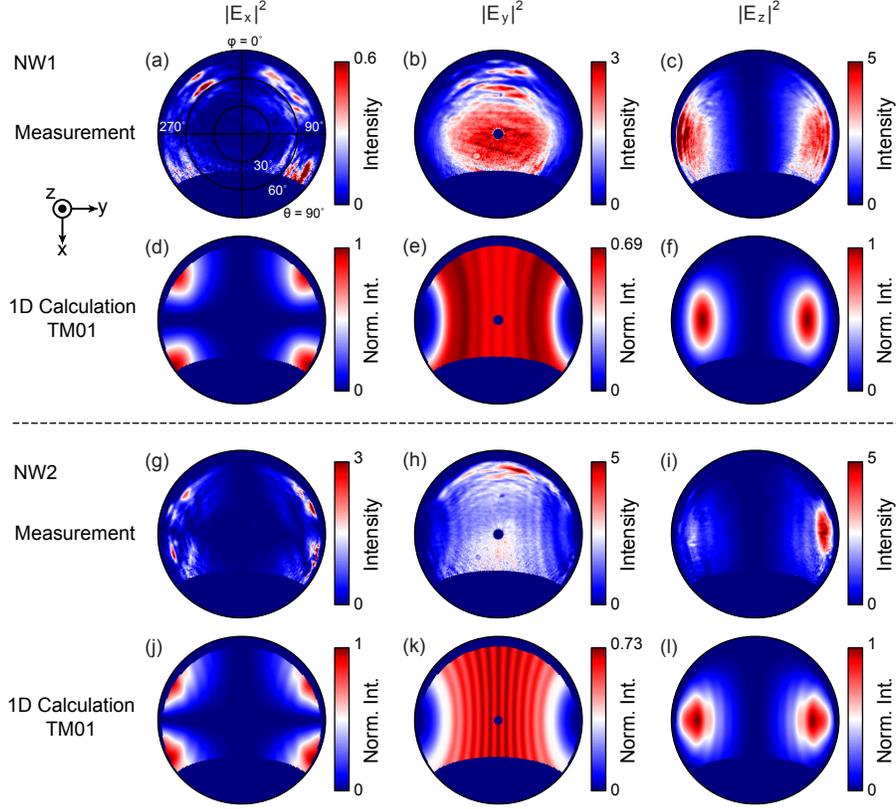


Figure 2: Measured (a-c, g-i) and calculated (d-f, j-l) angular emission distributions of the Cartesian field intensities at $\lambda_0 = 850$ nm for NW1 (a-f) and NW2 (g-l), as a function of azimuthal (φ) and zenithal (θ) angles. The patterns were measured and calculated for central excitation of the nanowires. (a, d, g, j) show the intensity of the E_x field component, (b, e, h, k) the intensity of E_y , and (c, f, i, l) the intensity of E_z (the coordinate system is shown in the top left). The calculations for each wire determine the far field emission profiles for the TM01 mode. We use the full range of the color scale for each panel, but the intensity scales for all three field components of each nanowire are normalized to the maximum value (E_x and E_z are equal). The measured intensities are given in 10^6 counts $\text{sr}^{-1} \text{s}^{-1}$.

113 the TM01 mode as being the dominant contribution, as shown in Figure 2. For NW1
 114 we compare the measurements (Figures 2(a-c)) to the 1D calculation for the TM01 mode
 115 (Figures 2(d-f)), displaying the Cartesian electric field intensities $|E_x|^2$, $|E_y|^2$ and $|E_z|^2$ as
 116 a function of azimuthal (φ) and zenithal (θ) angles. The field orientations are indicated by
 117 the coordinate system at the left, and the wires are oriented along the y-axis. A wavevector
 118 of $k_y = 6.63 + i 1.19 \mu\text{m}^{-1}$ was used for the calculation, as determined from the dispersion
 119 relation and nanowire diameter. The dark blue regions around the edges of each image

120 correspond to the angles at which no light is collected by the mirror. The intensity scale
 121 is chosen so as to maximize the contrast in the color scale to better view the details of the
 122 features. In the case of the calculation, the intensities are normalized to the overall maximum
 123 value for each wire. We observe excellent qualitative agreement between measurement and
 124 calculation. For $|E_x|^2$ (Figures 2(a,d)) there are four bright features at large zenithal angles,
 125 while $|E_y|^2$ (Figures 2(b,e)) displays bright emission in the center of the mirror and $|E_z|^2$
 126 (Figures 2(c,f)) exhibits two lobes to the left and right of the polar image, in the directions
 127 of the end facets of the nanowire.

128 For NW2 we also observe very similar features for both measurements (Figures 2(g-i)) and
 129 calculations (Figures 2(j-l)). A wavevector of $k_y = 8.00 + i 0.50 \mu\text{m}^{-1}$ was used for the
 130 calculation in this case. For $|E_x|^2$ (Figures 2(g,j)) we observe four features at slightly higher
 131 zenithal angles than for NW1, at the corners of the angular range. $|E_y|^2$ (Figures 2(h,k))
 132 shows the brightest intensity in the center, as for NW1, but this time we can also see
 133 intensity fringes along the vertical direction, which are due to interference between the
 134 emission from the nanowire end facets. The fringes are clearly visible in the experiment,
 135 but with lower contrast than in the calculations, which we attribute to imperfections in the
 136 mirror and the nanowire end facets, and to limitations on the angular resolution. Finally,
 137 $|E_z|^2$ (Figures 2(i,l)) again displays two lobes to the left and right, but at higher angles than
 138 for NW1, similarly to the behavior of $|E_x|^2$. In the experiment, the two lobes are asymmetric,
 139 which we attribute to the slight tapering of the wire. The emission is brighter in the direction
 140 of the thinner side. Comparing the relative intensities of calculations and measurements for
 141 both nanowires, we find that $|E_x|^2$ is weaker in the measurements than in the calculations.
 142 We ascribe this discrepancy to a lower collection efficiency at the edges of the mirror where
 143 the $|E_x|^2$ component is strongest.

144 Even though the two nanowires have quite different diameters, in both cases we can
 145 clearly recognize very similar polarized field distributions that show excellent qualitative
 146 agreement with calculations for the TM01 mode. From this we conclude that the emission

147 behavior of both NW1 and NW2 is dominated by the TM01 mode. The fundamental HE11
148 mode does not appear to play a major role in our case. This has been observed previously²²
149 and is ascribed to the fact that the fields are localized more outside of the wire for the HE11
150 mode than for the TM01 mode, which shows relatively more intensity in the center of the
151 wire, allowing it to couple more strongly. For comparison, we include calculations of the far
152 field emission profiles of the HE11 mode for central excitation of the thin and thick nanowires
153 in Figure S4 of the Supporting Information. Examining all field components, we find there
154 is much better agreement with the TM01 mode than with the HE11 mode. This confirms
155 that the CL emission couples most efficiently to the TM01 mode. Differences between the
156 two wires are expected, however, because for NW1 the mode is leaky ($Re(k_y) = 6.63 < k_0 =$
157 $7.39 \mu\text{m}^{-1}$), while for NW2 it is guided ($Re(k_y) = 8.00 > k_0 = 7.39 \mu\text{m}^{-1}$). To support the
158 data, we show polarization-resolved measurements for an additional thin and thick nanowire
159 in Figure S2 of the Supporting Information, which exhibit the same type of features for all
160 three field components as the results shown here.

161 **Directional emission**

162 Next, we study the directional behavior of the nanowire emission for excitation off-center,
163 near the end facets of the wires, observing a distinct difference in the directionality of the
164 emission between NW1 and NW2, as shown in Figure 3. We compare the total intensity that
165 both wires emit as a function of the azimuthal and zenithal angles in the case of measure-
166 ments, calculations and simulations, for excitation at the left edge, center, and right edge.
167 The edge excitation is always a few hundred nm away from the end facet, with the exact
168 positions shown by the dashed lines in Figure 4. We compare the total intensity measure-
169 ments to calculations using the dominant TM01 mode. Polarimetry measurements for select
170 positions near the end facets (not shown here) display the same characteristic features as in
171 Figure 2, so we do not observe a transition to a different mode at the edges. The measured

172 intensities differ between the wires and excitation positions, which we attribute to variations
173 in local material quality and size of the interaction volume (due to tapering and different
174 diameters). Here the total emission intensity is brighter when exciting the thicker ends.
175 For the measurements on NW1, central excitation (Figure 3(b)) results in two symmetric
176 lobes of higher intensity to the left and right, while excitation at the left edge (Figure 3(a))
177 leads to directional emission to the right side and excitation on the right (Figure 3(c)) leads
178 to emission towards the left side. 1D calculations of the total emission intensity from the
179 leaky TM01 mode qualitatively reproduce the emission behavior for excitation in the center
180 and 300 nm from the end facets (Figures 3(d-f)). In the measurements, the electron beam
181 excitation at the edges was $\sim 300\text{--}500$ nm from the end facets.

182 We attribute the discrepancies in the shape of the emission patterns between measure-
183 ments and calculations to the fact that the excitation volume can be much larger than the
184 electron beam width (up to a few hundred nm). This is due to electron scattering, secondary
185 electron generation, carrier diffusion, and photon recycling, which can play a large role in
186 such a direct band gap material.^{42,49,50} A large majority of excitations occur very close to
187 the point of impact, but light generation will cover a larger area. The overall spatial reso-
188 lution is determined by a convolution of all of these effects and will depend on the material
189 properties. For such strongly luminescent materials as GaAs the resolution is not as good as
190 the electron beam size but better than the full interaction volume. This is different from the
191 calculations, which assume a point-source. The presence of the thin holey carbon substrate,
192 which is not taken into account in the calculation, can also affect the emission, as we will
193 now show for NW2.

194 The measurements on the thicker NW2 (Figures 3(g-i)) show the opposite directionality
195 to that of NW1. Excitation at the left edge leads to emission towards the left, while excitation
196 at the right edge produces emission towards the right. The excitation positions were 700 nm
197 (left) and 400 nm (right) away from the end facets (see also the dashed lines in Figure 4(b)).
198 For central excitation we observe asymmetrical emission, as was the case for Figure 2(i),

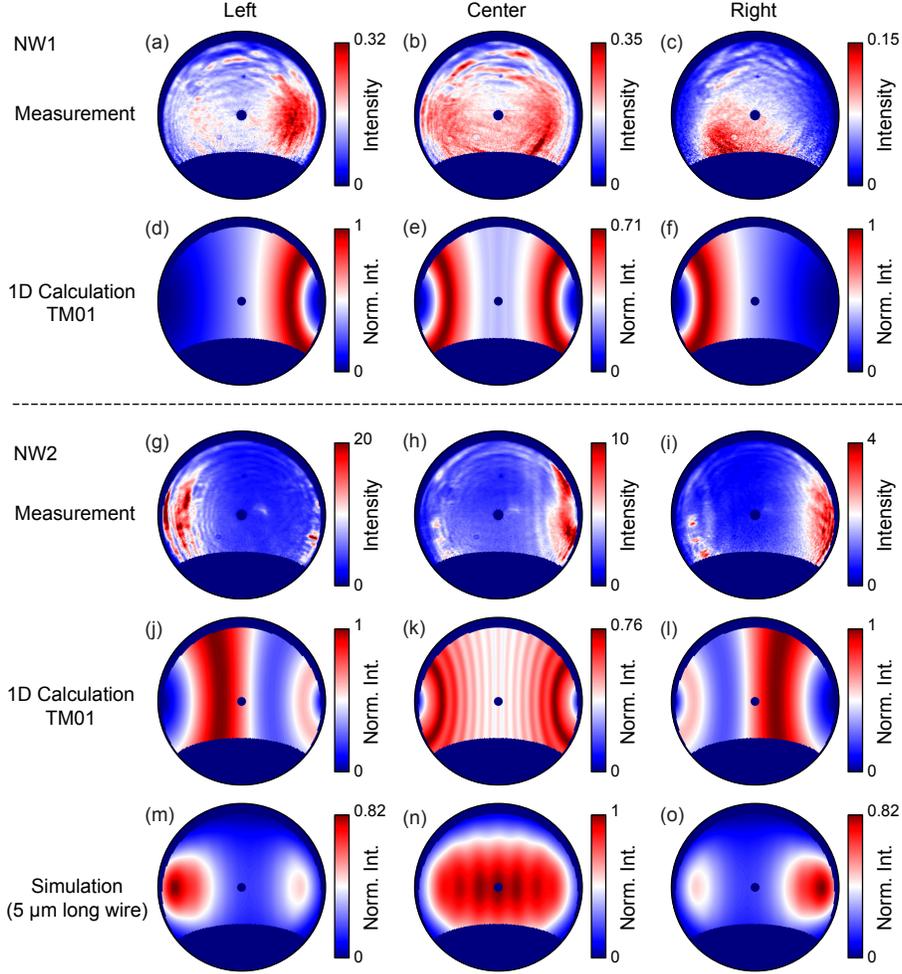


Figure 3: Measured (a-c, g-i), calculated (d-f, j-l) and simulated (m-o) angular emission distributions of the total intensity at $\lambda_0 = 850$ nm for NW1 (a-f) and NW2 (g-o). The patterns were measured and calculated for excitation at the left (a, d, g, j, m), center (b, e, h, k, n) and right (c, f, i, l, o) of the nanowires (see Figure 4 for positions). The calculations and simulations for each wire have been normalized to their maximum. The measured intensities are given in 10^6 counts $\text{sr}^{-1} \text{s}^{-1}$. The 1D calculation uses the same wire lengths as in the experiment (7.9 and 12 μm), but due to computational constraints the simulated NW2 is shorter (5 μm).

199 which we again attribute to the tapering of the wire that creates an inherent asymmetry in
 200 the wire and its emission properties. The tapering affects the leaky mode in the thin wire less
 201 since radiation is emitted continuously as the mode propagates along the wire. The thicker
 202 NW2 on the other hand, supports a guided mode, so light mostly escapes from the end
 203 facets and has a longer propagation length, traveling through the wire for multiple round-

trips. Since the modal properties are very sensitive to the diameter, the gradual variations along the length of the wire will affect the light more strongly.

We first compare the measurements to the 1D calculations of the (guided) TM₀₁ mode, which do not directly take the substrate into account (Figures 3(j-l)). We represent absorption at the band edge and losses into the substrate by an imaginary part of k_y of $0.50 i \mu\text{m}^{-1}$; this leads to an effective absorption length of $2 \mu\text{m}$, much larger than the TM₀₁ wavelength ($\sim 220 \text{ nm}$), but shorter than the NW length, thus limiting mode bouncing at the NW edges. We find that for excitation near the edges (500 nm away from the end facet, similarly to the measurement) there is a maximum in emission to the same side as in the measurements, with a weaker feature in the opposite direction. We note that in the measurements of NW2 there is also a region of higher intensity to the opposite side of the dominant emission. For central excitation, we observe quite good qualitative agreement between experiment and calculation, taking into account the asymmetry we attribute to tapering of the wire. As shown in Figure 3(k), interference fringes from the emission of both facets are expected for this long wire. These are also faintly visible in Figure 3(h).

To get a better measure for the effect of the substrate, we perform numerical simulations using COMSOL (see Methods for more details) on a 180 nm thick and $5 \mu\text{m}$ long wire on a semi-infinite carbon substrate (Figures 3(m-o)). Due to computational constraints we did not simulate a $12 \mu\text{m}$ long wire nor the extremely thin substrate. The simulations, however, do show good qualitative agreement with the experiment and provide insight into the role of the substrate on the emission behavior. Central excitation leads to a symmetric emission profile with highest intensity in the central region and interference fringes that are less distinct than for the 1D calculation. Excitation at the edges (500 nm from the end facets) shows emission profiles in good qualitative agreement with the measurements. There is a bright feature on the same side at high angles and a weaker spot on the opposite side. Both the 1D calculations and the simulations predict the measured directionality, which is completely opposite to the behavior of NW1. The features measured for edge excitation

231 closely resemble the simulation, while there is better agreement with the 1D calculation for
232 central excitation. As the substrate is very thin, we can expect it to have a smaller effect
233 than in the simulation that was performed for a semi-infinite substrate. The importance
234 of the substrate as an additional loss channel does not play a large role in the case of
235 the leaky mode ($k_y < k_0$) as there is already a strong inherent leakage. For the thicker
236 wire, simulations without substrate show an emission directionality that is more strongly
237 dependent on excitation position and near the edges becomes opposite to that observed in
238 the measurements (Figure S4 and Figure S5 in the Supporting Information). We conclude
239 that both the guided behavior of the TM01 mode and the additional loss channel due to the
240 substrate play a role in determining the directional emission behavior of the thick nanowire.

241 The changing directionality observed in the measurements and calculations may be un-
242 derstood in an intuitive manner, when examining the differences between leaky and guided
243 modes. For the leaky mode, leakage of the light along the nanowire dominates the emission.
244 When exciting close to an end facet, light propagating to the edge will partially reflect back,
245 while light going to the opposite side will propagate longer and thus leak out more, leading
246 to a majority of the emission into the opposite direction. For the guided mode, emission
247 from the end facets dominates, while there is loss to absorption into the substrate for light
248 propagating along the wire, so more light will scatter out from the closest edge than from
249 the far edge.

250 We can study the directional behavior of the emission as a function of the excitation
251 position more closely, taking advantage of the high spatial resolution of CL. As discussed
252 previously, the resolution is not limited to the electron beam size, but still remains sub-
253 wavelength. Figure 4 shows the emission directionality for both wires when scanning the
254 beam along their length. We determine a left-to-right ratio $L-R/L+R$ by averaging the to-
255 tal intensity over all zenithal angles in 60° azimuthal wedges on the left and right sides, as
256 these correspond to the regions of highest intensity features. The gray bands correspond
257 to positions that are outside the wires and the dashed lines indicate the positions of the

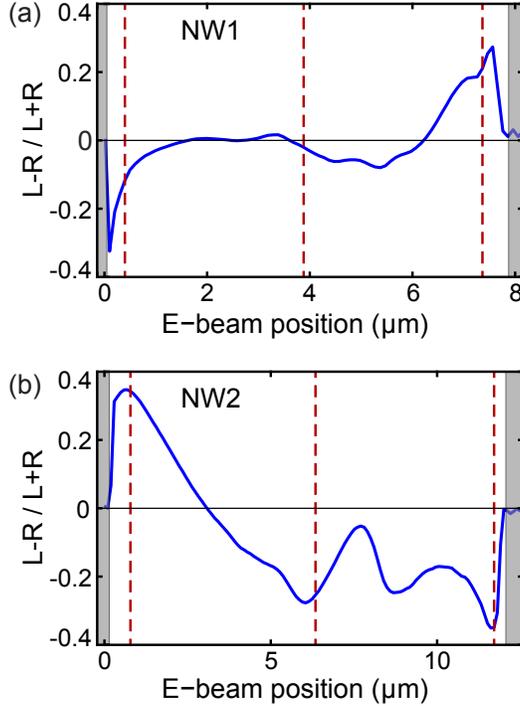


Figure 4: Ratio of the left-to-right directional emission for NW1 (a) and NW2 (b), showing the ratio $(L-R)/(L+R)$ as a function of the electron beam position as it scans along the wire. The gray bands indicate positions that are not on the wire, while the red dashed lines indicate the positions of the left, center and right measurements shown in Figure 3. The leftwards and rightwards directional intensities were determined by averaging the total intensity over all zenithal angles in 60° azimuthal wedges ($\varphi=240-300^\circ$ for left and $\varphi=60-120^\circ$ for right).

258 measurements in Figure 3.

259 Comparing NW1 (Figure 4(a)) to NW2 (Figure 4(b)), we observe that there is no
 260 left/right directionality at the very edges for both wires, but that close to the edges the
 261 left-to-right ratio reaches a maximum which is reversed for the two wires, as expected from
 262 Figure 3. Figure S6 in the Supporting Information shows the left-to-right ratio for simula-
 263 tions of a thick wire, which also exhibit a maximum close to the end facet. Additionally, the
 264 simulations with substrate show better agreement with the measurements than the simula-
 265 tions without. The directionality we observe results from interference of waves propagating
 266 back and forth in the nanowire, which is dependent on the reflection at the end facets, ab-
 267 sorption and leakage during propagation but also on the excitation position. This peak in

268 emission close to the end facet is convoluted with the interaction volume of the electrons
269 with the material. At the very edge we do not excite as large a region, which contributes to
270 the decrease in intensity and directionality we observe.

271 The change in emission directionality observed here is consistent with the additional thin
272 and thick nanowires examined in the Supporting Information (Figure S3). Comparing all
273 nanowires, the thickest diameter for the leaky behavior in the thinner wires is 145 nm, while
274 the thinnest diameter for the guided behavior in the thicker wires is 170nm. This indicates
275 that the transition in emission directionality should occur for nanowire diameters in the
276 range of 145-170 nm.

277 **Conclusion**

278 In conclusion, we have demonstrated that cathodoluminescence emission from GaAs nanowires
279 is strongly directional and depends on the nanowire diameter. The emission excited by the
280 electron beam couples to waveguide modes that determine the polarization and angular dis-
281 tribution of the outcoupled radiation. These waveguide modes are very sensitive to wire
282 diameter, especially as they change in nature from leaky to guided when crossing the light
283 line in air. Polarization-resolved measurements show that the TM₀₁ mode dominates the
284 emission from both nanowires. The thin wire supports a leaky TM₀₁ mode, which displays
285 emission in the direction opposite from the excited edge, while the thick wire supports a
286 guided TM₀₁ mode that exhibits emission in the same direction. The emission direction-
287 ality switches for nanowire diameters in the range of 145-170 nm. Both the leaky/guided
288 nature of the mode and the presence of the substrate play an important role in determining
289 the emission directionality. Cathodoluminescence polarimetry proves to be a powerful tech-
290 nique to study the angular- and polarization-dependent emission properties of semiconductor
291 nanowires or other nanostructures, with a subwavelength excitation resolution.

292 **Methods**

293 **Sample fabrication**

294 The GaAs nanowires were grown on a Si(111) undoped wafer via a Ga-assisted method in
295 a DCA P600 solid-source MBE machine.^{43,44} Typical growth parameters are as follows: a
296 Ga rate of 0.3 Å/s as flux of 2.5×10^{-6} torr, a substrate temperature of 640 °C, rotation of
297 the substrate at 7 r.p.m., and a V/III beam equivalent pressure ratio of 50. The nanowires
298 were removed from the silicon substrate in an isopropanol solution by ultrasonic bath for 1
299 minute. A few drops of the isopropanol solution containing nanowires were transferred to a
300 holey carbon TEM grid (Plano GmbH).

301 **CL measurements**

302 The measurements were performed in a FEI XL-30 SFEG (5 keV electron beam, ~ 0.1 nA
303 current) equipped with a home-built CL system.^{33,45,46} The emission excited by the electron
304 beam is collected by an aluminium paraboloid mirror and directed to an optical setup. We
305 measure either the spectrum using a liquid-nitrogen-cooled back-illuminated silicon CCD
306 array (Princeton Instruments Spec-10 100B), or the angular emission profile using a Peltier-
307 cooled back-illuminated 2D silicon CCD array (Princeton Instruments PIXIS 1024B).^{45,46}
308 Using a series of six measurements of the angular CL distribution with the 2D CCD array
309 in conjunction with a quarter-wave plate (QWP) and linear polarizer (LP) determines the
310 full emission polarization. Each measurement was taken for a different combination of QWP
311 and LP settings (horizontal, vertical, 45°, 135°, right- and left-handed circular). We correct
312 for the geometrical and polarization dependent response of the paraboloid mirror on the CL
313 emission that it redirects to the optical setup.⁴¹ A 40 nm bandpass color filter was used to
314 spectrally select the measured emission at $\lambda_0 = 850$ nm. Integration times of 0.1–1 s were
315 used depending on sample brightness. For every setting of the QWP and LP, we collected a
316 dark reference measurement where we blank the electron beam (using the same integration

317 time as for the corresponding CL measurement). This reference was subtracted from the
318 data in the post-processing stage. Possible sources of measurement errors include drift of
319 the electron beam, bleaching/contamination which leads to a reduction in CL signal, and
320 fluctuations in the current and/or the alignment of the mirror.

321 **FEM simulations**

322 The finite-element-method (FEM) simulations of the far field emission profiles of finite
323 nanowires were performed using the commercial software package COMSOL Multiphysics
324 v4.3b, using the same methods as in Ref. 27 and Ref. 23. For free-standing nanowires the
325 simulation space consisted of a circular cylinder of length L and diameter d that represents
326 the nanowire, enclosed in three concentric spheres of diameter $L + 2\lambda_0$, $L + 4\lambda_0$, and $L + 6\lambda_0$,
327 with their centers coinciding with that of the cylinder. The innermost two spheres were set
328 to be air (n_{air}), while the outermost layer was defined as a perfectly matched layer (PML)
329 to absorb all outgoing radiation and prevent reflections. The material constants of GaAs for
330 the cylinder were taken from Palik⁵¹ ($n_{GaAs} = 3.6$ at $\lambda_0 = 850$ nm). A tetrahedral mesh was
331 used, with maximum elements sizes (MES) of 25 nm in the domain of the cylinder and 160
332 nm for the air domains. The maximum element growth rate was set to 1.35 for all of the
333 domains.

334 For nanowires on top of a carbon substrate the geometry is modified as follows. The
335 three concentric spheres of the same diameter are divided into two semi-spherical layered
336 domains through a plane that contains the cylinder axis and the cylinder is then shifted by
337 $d/2$ from its original position in order to be placed on top of one of the new semi-spherical
338 spaces, which we refer to as the substrate. The substrate was set to be amorphous carbon
339 ($n_C = 1.987 + i 0.83$ at $\lambda_0 = 850$ nm)⁵² and the rest was set to be air, except for the GaAs
340 cylinder. As the space was divided into two different media, the material properties of the
341 outermost PML must be the same as the adjacent medium. The MES of the tetrahedral
342 mesh was 25 nm for the cylinder, 160 nm for the air and 90 nm for the substrate. The

343 maximum element growth rate was 1.35, the same as for the free-standing nanowires.

344 Simulations were highly memory-demanding; in the case of the nanowires of length $L =$
345 $5 \mu\text{m}$ on top of the substrate, the calculations need ~ 400 GB. Post-processing calculations
346 were used to determine the total radiated power at the inner spherical boundary Σ_{int} ,
347 defined by:

$$P = \int_{\Sigma_{int}} \langle \mathbf{S} \rangle \cdot \mathbf{n} dS \quad (1)$$

348 where \mathbf{n} is the outward normal unit-vector to the surface.

349 **Supporting Information Available**

350 The Supporting Information contains data from an additional two measured nanowires,
351 calculated emission profiles for the HE11 mode, as well as simulations that compare the
352 emission behavior with and without the substrate.

353 This material is available free of charge via the Internet at <http://pubs.acs.org/>.

354 **Author Information**

355 **Corresponding Authors**

356 *E-mail (A. Polman): polman@amolf.nl.

357 **Notes**

358 A.P. is co-founder and co-owner of Delmic BV, a startup company developing a commercial
359 product based on the cathodoluminescence system that was used in this work.

Acknowledgement

The authors acknowledge Clara Osorio, Mark Knight and Toon Coenen for fruitful discussions. We also thank Mohammad Ramezani, Gözde Tütüncüoğlu, Federico Matteini, and Anna Fontcuberta i Morral for providing the GaAs sample. This work is part of the “Stichting voor Fundamenteel Onderzoek der Materie (FOM)” as well as the Dutch technology foundation STW, which are financially supported by the “Nederlandse Organisatie voor Wetenschappelijk Onderzoek (NWO)” and the Dutch Ministry of Economic Affairs. It is also part of NanoNextNL, a nanotechnology program funded by the Dutch Ministry of Economic Affairs, part of an industrial partnership program between Philips and FOM, and is supported by the European Research Council (ERC). The Spanish “Ministerio de Economía y Competitividad” is also acknowledged for financial support through the grants NANOPLAS+ (FIS2012-31070) and LENSBEAM (FIS2015-69295-C3-2-P).

References

- (1) Huang, M. H.; Mao, S.; Feick, H.; Yan, H.; Wu, Y.; Kind, H.; Weber, E.; Russo, R.; Yang, P. Room-temperature ultraviolet nanowire nanolasers. *Science* **2001**, *292*, 1897–1899.
- (2) Vanmaekelbergh, D.; van Vugt, L. K. ZnO nanowire lasers. *Nanoscale* **2011**, *3*, 2783–2800.
- (3) Röder, R.; Ploss, D.; Kriesch, A.; Buschlinger, R.; Geburt, S.; Peschel, U.; Ronning, C. Polarization features of optically pumped CdS nanowire lasers. *J. Phys. D: Appl. Phys.* **2014**, *47*, 394012.
- (4) Haraguchi, K.; Katsuyama, T.; Hiruma, K. Polarization dependence of light emitted from GaAs p-n junctions in quantum wire crystals. *J. Appl. Phys.* **1994**, *75*, 4220–4225.

- 383 (5) Svensson, C. P. T.; Mårtensson, T.; Trägårdh, J.; Larsson, C.; Rask, M.; Hessman, D.;
384 Samuelson, L.; Ohlsson, J. Monolithic GaAs/InGaP nanowire light emitting diodes on
385 silicon. *Nanotechnology* **2008**, *19*, 305201.
- 386 (6) Wallentin, J.; Anttu, N.; Asoli, D.; Huffman, M.; Åberg, I.; Magnusson, M. H.;
387 Siefert, G.; Fuss-Kailuweit, P.; Dimroth, F.; Witzigmann, B.; Xu, H. Q.; Samuelson, L.;
388 Deppert, K.; Borgström, M. T. InP Nanowire Array Solar Cell Achieving 13.8% Effi-
389 ciency by Exceeding the Ray Optics Limit. *Science* **2013**, *339*, 1057–1060.
- 390 (7) Cui, Y.; Wang, J.; Plissard, S. R.; Cavalli, A.; Vu, T. T. T.; van Veldhoven, R. P. J.;
391 Gao, L.; Trainor, M.; Verheijen, M. A.; Haverkort, J. E. M.; Bakkers, E. P. A. M.
392 Efficiency enhancement of InP nanowire solar cells by surface cleaning. *Nano Lett.*
393 **2013**, *13*, 4113–7.
- 394 (8) Krogstrup, P.; Jørgensen, H. I.; Heiss, M.; Demichel, O.; Holm, J. V.; Aagesen, M.;
395 Nygard, J.; Fontcuberta i Morral, A. Single-nanowire solar cells beyond the Shockley-
396 Queisser limit. *Nat. Photonics* **2013**, *7*, 306–310.
- 397 (9) Muskens, O. L.; Gómez Rivas, J.; Algra, R. E.; Bakkers, E. P. A. M.; Lagendijk, A.
398 Design of light scattering in nanowire materials for photovoltaic applications. *Nano*
399 *Lett.* **2008**, *8*, 2638–2642.
- 400 (10) Claudon, J.; Bleuse, J.; Malik, N. S.; Bazin, M.; Jaffrennou, P.; Gregersen, N.;
401 Sauvan, C.; Lalanne, P.; Gérard, J.-M. A highly efficient single-photon source based on
402 a quantum dot in a photonic nanowire. *Nat. Photonics* **2010**, *4*, 174–177.
- 403 (11) Heiss, M. et al. Self-assembled quantum dots in a nanowire system for quantum pho-
404 tonics. *Nat. Mater.* **2013**, *12*, 439–444.
- 405 (12) Bulgarini, G.; Reimer, M. E.; Bouwes Bavinck, M.; Jöns, K. D.; Dalacu, D.; Poole, P. J.;
406 Bakkers, E. P. A. M.; Zwiller, V. Nanowire waveguides launching single photons in a
407 gaussian mode for ideal fiber coupling. *Nano Lett.* **2014**,

- 408 (13) Reimer, M. E.; Bulgarini, G.; Akopian, N.; Hocevar, M.; Bouwes Bavinck, M.; Verhei-
409 jen, M. A.; Bakkers, E. P. A. M.; Kouwenhoven, L. P.; Zwiller, V. Bright single-photon
410 sources in bottom-up tailored nanowires. *Nat. Commun.* **2012**, *3*, 737.
- 411 (14) Wang, J.; Gudiksen, M. S.; Duan, X.; Cui, Y.; Lieber, C. M. Highly polarized pho-
412 toluminescence and photodetection from single indium phosphide nanowires. *Science*
413 **2001**, *293*, 1455.
- 414 (15) Paniagua-Domínguez, R.; Abujetas, D. R.; Sánchez-Gil, J. A. Ultra low-loss, isotropic
415 optical negative-index metamaterial based on hybrid metal-semiconductor nanowires.
416 *Sci. Rep.* **2013**, *3*, 1507.
- 417 (16) Ramezani, M.; Casadei, A.; Grzela, G.; Matteini, F.; Tütüncüoğlu, G.; Ruffer, D.;
418 Fontcuberta i Morral, A.; Gómez-Rivas, J. Hybrid semiconductor nanowire-metallic
419 Yagi-Uda antennas. *Nano Lett.* **2015**, *15*, 4889–4895.
- 420 (17) Yan, R.; Gargas, D.; Yang, P. Nanowire photonics. *Nat. Photonics* **2009**, *3*, 569–576.
- 421 (18) Duan, X.; Huang, Y.; Cui, Y.; Wang, J.; Lieber, C. M. Indium phosphide nanowires as
422 building blocks for nanoscale electronic and optoelectronic devices. *Nature* **2001**, *409*,
423 66–69.
- 424 (19) Gudiksen, M. S.; Lauhon, L. J.; Wang, J.; Smith, D. C.; Lieber, C. M. Growth of
425 nanowire superlattice structures for nanoscale photonics and electronics. *Nature* **2002**,
426 *415*, 617–620.
- 427 (20) Thelander, C.; Agarwal, P.; Brongersma, S.; Eymery, J.; Feiner, L.; Forchel, A.;
428 Scheffler, M.; Riess, W.; Ohlsson, B.; Gösele, U.; Samuelson, L. Nanowire-based one-
429 dimensional electronics. *Mater. Today* **2006**, *9*, 28 – 35.
- 430 (21) Motohisa, J.; Kohashi, Y.; Maeda, S. Far-field emission patterns of nanowire light-
431 emitting diodes. *Nano Lett.* **2014**, *14*, 3653–60.

- 432 (22) Grzela, G.; Paniagua-Domínguez, R.; Barten, T.; Fontana, Y.; Sánchez-Gil, J. A.;
433 Gómez Rivas, J. Nanowire antenna emission. *Nano Lett.* **2012**, *12*, 5481–5486.
- 434 (23) van Dam, D.; Abujetas, D. R.; Paniagua-Domínguez, R.; Sánchez-Gil, J. A.; Bakkers, E.
435 P. A. M.; Haverkort, J. E. M.; Gómez Rivas, J. Directional and polarized emission from
436 nanowire arrays. *Nano Lett.* **2015**, *15*, 4557–4563.
- 437 (24) Maslov, A. V.; Ning, C. Z. Far-field emission of a semiconductor nanowire laser. *Opt.*
438 *Lett.* **2004**, *29*, 572–574.
- 439 (25) Maslov, A. V.; Bakunov, M. I.; Ning, C. Z. Distribution of optical emission between
440 guided modes and free space in a semiconductor nanowire. *J. Appl. Phys.* **2006**, *99*,
441 024314.
- 442 (26) Cao, L.; S., W. J.; Park, J.; Schuller, J. A.; Clemens, B. M.; Brongersma, M. L.
443 Engineering light absorption in semiconductor nanowire devices. *Nat. Mater.* **2009**, *8*,
444 643–647.
- 445 (27) Paniagua-Domínguez, R.; Grzela, G.; Gómez Rivas, J.; Sánchez-Gil, J. Enhanced and
446 directional emission of semiconductor nanowires tailored through leaky/guided modes.
447 *Nanoscale* **2013**, *5*, 10582–10590.
- 448 (28) Claudon, J.; Gregersen, N.; Lalanne, P.; Gérard, J.-M. Harnessing light with photonic
449 nanowires: fundamentals and applications to quantum optics. *ChemPhysChem* **2013**,
450 *14*, 2393–2402.
- 451 (29) Abujetas, D. R.; Paniagua-Domínguez, R.; Sánchez-Gil, J. A. Unraveling the janus role
452 of Mie resonances and leaky/guided modes in semiconductor nanowire absorption for
453 enhanced light harvesting. *ACS Photonics* **2015**, *2*, 921–929.
- 454 (30) Traviss, D. J.; Schmidt, M. K.; Aizpurua, J.; Muskens, O. L. Antenna resonances in
455 low aspect ratio semiconductor nanowires. *Opt. Express* **2015**, *23*, 22771–22787.

- 456 (31) García de Abajo, F. J. Optical excitations in electron microscopy. *Rev. Mod. Phys.*
457 **2010**, *82*, 209–275.
- 458 (32) Adamo, G.; Ou, J. Y.; So, J. S.; Jenkins, S. D.; De Angelis, F.; MacDonald, K. F.;
459 Di Fabrizio, E.; Ruostekoski, J.; Zheludev, N. I. Electron-beam-driven collective-mode
460 metamaterial light source. *Phys. Rev. Lett.* **2012**, *109*, 217401.
- 461 (33) Sapienza, R.; Coenen, T.; Renger, J.; Kuttge, M.; van Hulst, N. F.; Polman, A. Deep-
462 subwavelength imaging of the modal dispersion of light. *Nat. Mater.* **2012**, *11*, 781–787.
- 463 (34) Bashevoy, M. V.; Jonsson, F.; MacDonald, K. F.; Chen, Y.; Zheludev, N. I. Hyper-
464 spectral imaging of plasmonic nanostructures with nanoscale resolution. *Opt. Express*
465 **2007**, *15*, 11313–11320.
- 466 (35) Zhu, X. L.; Ma, J. S., Y. Zhang; Xu, X. F., J. Wu; Zhang, Y.; Han, X. B.; Fu, Q.;
467 Liao, Z. M.; Chen, L.; Yu, D. P. Confined three-dimensional plasmon modes inside a
468 ring-shaped nanocavity on a silver film imaged by cathodoluminescence microscopy.
469 *Phys. Rev. Lett.* **2010**, *105*, 127402.
- 470 (36) Takeuchi, K.; Yamamoto, N. Visualization of surface plasmon polariton waves in two-
471 dimensional plasmonic crystal by cathodoluminescence. *Opt. Express* **2011**, *19*, 12365–
472 12374.
- 473 (37) Coenen, T.; van de Groep, J.; Polman, A. Resonant Modes of Single Silicon Nanocavi-
474 ties Excited by Electron Irradiation. *ACS Nano* **2013**, *7*, 1689–1698.
- 475 (38) Yacobi, B. G.; Holt, D. B. *Cathodoluminescence microscopy of inorganic solids*; Springer
476 Science & Business Media, 1990.
- 477 (39) Edwards, P. R.; Martin, R. W. Cathodoluminescence nano-characterization of semicon-
478 ductors. *Semicond. Sci. Technol.* **2011**, *26*, 064005.

- 479 (40) Thonke, K.; Tischer, I.; Hocker, M.; Schirra, M.; Fujan, K.; Wiedenmann, M.; Schnei-
480 der, R.; Frey, M.; Feneberg, M. Nanoscale characterisation of semiconductors by
481 cathodoluminescence. *IOP Conf. Ser.: Mater. Sci. Eng.* **2014**, *55*, 012018.
- 482 (41) Osorio, C. I.; Coenen, T.; Brenny, B. J. M.; Polman, A.; Koenderink, A. F. Angle-
483 resolved cathodoluminescence imaging polarimetry. *ACS Photonics* **2016**, *3*, 147–154.
- 484 (42) Brenny, B. J. M.; van Dam, D.; Osorio, C. I.; Gómez Rivas, J.; Polman, A. Az-
485 imuthally polarized cathodoluminescence from InP nanowires. *Appl. Phys. Lett.* **2015**,
486 *107*, 201110.
- 487 (43) Russo-Averchi, E.; Heiss, M.; Michelet, L.; Krogstrup, P.; Nygard, J.; Magen, C.;
488 Morante, J. R.; Uccelli, E.; Arbiol, J.; Fontcuberta i Morral, A. Suppression of three
489 dimensional twinning for a 100% yield of vertical GaAs nanowires on silicon. *Nanoscale*
490 **2012**, *4*, 1486–1490.
- 491 (44) Matteini, F.; Tütüncüoğlu, G.; Potts, H.; Jabeen, F.; Fontcuberta i Morral, A. Wetting
492 of Ga on SiO_x and its impact on GaAs nanowire growth on silicon. *Cryst. Growth. Des.*
493 **2015**, *15*, 3105–3109.
- 494 (45) Coenen, T.; Vesseur, E. J. R.; Polman, A.; Koenderink, A. F. Directional emission
495 from plasmonic Yagi Uda antennas probed by angle-resolved cathodoluminescence spec-
496 troscopy. *Nano Lett.* **2011**, *11*, 3779–3784.
- 497 (46) Coenen, T.; Vesseur, E. J. R.; Polman, A. Angle-resolved cathodoluminescence spec-
498 troscopy. *Appl. Phys. Lett.* **2011**, *99*, 143103.
- 499 (47) Born, M.; Wolf, E. *Principles of Optics: Electromagnetic Theory of Propagation, In-*
500 *terference and Diffraction of Light, 7th edition*; Cambridge University Press, 1997.
- 501 (48) Stratton, J. A. *Electromagnetic theory*; Read Books, 1941.

- 502 (49) Haegel, N. M.; Mills, T. J.; Talmadge, M.; Scandrett, C.; Frenzen, C. L.; Yoon, H.;
503 Fetzer, C. M.; King, R. R. Direct imaging of anisotropic minority-carrier diffusion in
504 ordered GaInP. *J. Appl. Phys.* **2009**, *105*, 023711.
- 505 (50) Demers, H.; Poirier-Demers, N.; Couture, A. R.; Joly, D.; Guilmain, M.; de Jonge, N.;
506 Drouin, D. Three-dimensional electron microscopy simulation with the CASINO Monte
507 Carlo software. *Scanning* **2011**, *33*, 135–146.
- 508 (51) Palik, E. D. *Handbook of optical constants of solids*; Academic press, 1998; Vol. 3.
- 509 (52) Hagemann, H.-J.; Gudat, W.; Kunz, C. Optical constants from the far infrared to the
510 x-ray region: Mg, Al, Cu, Ag, Au, Bi, C, and Al₂O₃. *JOSA* **1975**, *65*, 742–744.

511 **Graphical TOC Entry**

512

