# In-situ LIBS and NRA deuterium retention study in porous and compact W-O coatings loaded by Magnum-PSI

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# Abstract

Present study investigated the applicability of in-situ laser induced breakdown spectroscopy (LIBS) for deuterium retention measurements in tungsten oxide coatings which were exposed to a Gaussian beam of deuterium plasma in the Magnum-PSI linear plasma device. The deuterium line intensities determined by LIBS were compared with the deuterium content measured by nuclear reaction analysis (NRA).

Both LIBS and NRA results showed that higher deuterium retention was achieved in the coating region corresponding to the periphery of the plasma beam. This decreasing deuterium retention in the central region can be attributed to higher surface temperature. At the same time, the deuterium retention in different coating types assessed by D/H and D/Mo ratio of LIBS was markedly different from the retention determined by NRA. Porous W-O coatings had the highest deuterium retention according to NRA while the D/H and D/Mo intensity ratio obtained by LIBS was an order of magnitude smaller when compared with other coatings. The deuterium retention in compact W-O coating and thick W coating was almost the same and LIBS D/H ratios were also comparable for these coatings. The results demonstrate the LIBS applicability and its limits in different coating types.

*Keywords:* ITER relevant tungsten coatings, deuterium retention, plasma exposure in Magnum-PSI plasma, in-situ NRA, in-situ LIBS

## 1. Introduction

Successful exploitation of the International Thermonuclear Experimental Reactor (ITER) requires the knowledge of the fuel retention in the reactor walls [1–3]. In order to comply to the safety regulations, the total retained tritium (T) in the reactor vessel should be monitored to assure that the amount of tritium in the ITER vessel does not exceed the limit of 700 g. ITER will use two hydrogen isotopes, deuterium D and tritium T to fuel the fusion reaction. The fuel retention in ITER divertor walls subjected to plasma fluxes up to  $10^{24}$  ions/m<sup>2</sup>s can currently be systematically studied only in the linear plasma devices with D plasma [4,5]. The retention is strongly influenced by the properties of wall materials, i.e. temperature, structure and elemental composition. ITER divertor walls are made from tungsten but during plasma exposure of the wall erosion and redeposition of wall material may change the surface material composition and deuterium retention [6–8]. It has been shown that D retention in coatings which simulate these layers increases markedly in polycrystalline, nanocrystalline and amorphous tungsten coatings [9–11]. Furthermore, the D retention is influenced by the co-deposition of additional elements, for example oxygen which is present in plasma as an impurity [12,13].

Laser Induced Breakdown Spectroscopy (LIBS) is a method where a short but powerful laser pulse is used to ablate a tiny amount of material from the surface, which results in the formation of a plasma plume of wall material. The spectrum of the light emitted from the plasma plume is used to determine the elemental composition of the ablated material. It has been demonstrated that LIBS can be used for remote in-situ determination of the hydrogen isotopes in the W based materials both in tokamaks [14–17] and linear plasma devices [18–22]. The determination of absolute value of D content requires extensive calibration with different coatings. The calibration can be made by comparing D intensity obtained by LIBS with the absolute D content obtained by other methods. One method which determines both the total amount of D and depth distribution is nuclear reaction analysis (NRA) [9,23–27].

Our previous ex-situ LIBS study [28] of D retention in nanostructured compact and porous W-O coatings loaded by Magnum-PSI plasma demonstrated the LIBS ability to obtain qualitative correspondence with D retention in the coatings. However, a disadvantage was that this study was made several months after D loading. The D intensity determined by LIBS was compared with the relative D content obtained by Secondary Ion Mass Spectroscopy (SIMS) and inherently the results were influenced by adsorbed water from surrounding atmosphere and possible outgassing of D. The H signal originating from the surface water vapor interferes with D signal which has detrimental effect on the detection limit of D. More accurate D analysis requires experiments which are made immediately after D loading without exposing the samples to ambient air. Furthermore, it is necessary to calibrate the LIBS signal with the absolute D composition. The ion beam diagnostics which is installed on the Target Exchange and Analysis Chamber (TEAC) [4] enables to use NRA consecutively with LIBS in Magnum-PSI.

The aim of this study was the comparison of retention properties of samples with ITER like Wall (ILW) coatings of different composition and structure using LIBS and direct comparison of LIBS results with results obtained with NRA. The investigated samples were compact and porous W-O coatings which were analogous with the coatings used in our previous study [28]. These samples were exposed to deuterium plasma in Magnum PSI. The D retention was investigated by in-situ LIBS and the results were compared with the D concentration determined by in-situ NRA. In addition, W-O coatings and W coating were compared to investigate the influence of oxygen on deuterium retention.

# 2. Experimental setup

#### 2.1. Preparation of coatings

The investigated porous and compact W-O coatings were produced in the Micro and Nano Structred Materials Laboratory Department of Energy Politecnico di Milano and the samples consist of Mo substrates of 30 mm diameter and 1 mm thickness coated by pulsed laser deposition. A ns laser pulse (wavelength 532 nm, fluence 15 J/cm<sup>2</sup>) was focused on a pure metallic W target in 70 Pa argon or helium atmosphere to obtain approximately 1  $\mu$ m thick porous and compact coatings [29,30]. A 150 nm adhesion interlayer of crystalline W was deposited at 10<sup>-3</sup> Pa base pressure before the deposition of the porous coating. According to earlier studies [30] the compact coatings with thickness of 1.4  $\mu$ m contained 17% oxygen while porous coatings with thickness of 1.6  $\mu$ m contained 50% oxygen.

In addition, a compact pure W coating was prepared by reactive high-power impulse magnetron sputtering (HIPIMS) method in National Institute for Laser, Plasma and Radiation Physics, Bucharest. The composition and thickness of the W coating was determined by Spectruma GDA 750 Glow Discharge Optical Emission Spectrometry (GDOES) and SEM. The thickness of this coating was approximately 10  $\mu$ m with the uppermost 1  $\mu$ m layer containing up to 20% of oxygen. The schematic buildup of the samples with coatings and interlayers is shown in figure 1.

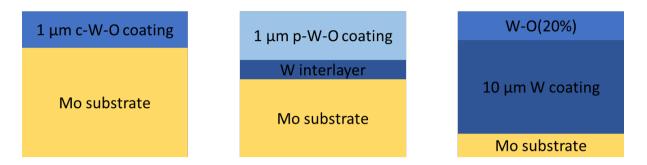


Figure 1. Schematic drawing of the sample composition for compact W-O (c-W-O) coating, porous W-O coating on W interlayer (p-W-O) and thick W coating with uppermost layer containing oxygen (W).

## 2.2. Deuterium loading in Magnum-PSI

The loading of deuterium into the coatings was made by the Magnum-PSI linear plasma generator of The Dutch Institute for Fundamental Energy Research (DIFFER) [5,31]. For all experiments, a D<sub>2</sub> flow of 5.0 slm was used at the plasma source current of 135 A and the plasma beam was confined by a magnetic field of 0.8 T. The Full Width at Half Maximum (FWHM) of the beam was 18 mm and 21 during plasma exposures of the porous and compact W-O coatings and W coating, respectively. The bias voltage of target was -40 V which results in an ion implantation energy of >15 eV. Thomson scattering [32] was used to measure the parameters of Magnum PSI plasma near the surface of the coating. The electron temperature for W-O coatings was 1.2-1.3 eV, electron density  $0.4 \cdot 10^{20}$  m<sup>-3</sup> ( $0.2 \cdot 10^{20}$  m<sup>-3</sup> for W) and particle flux  $2.5 \cdot 10^{23}$  m<sup>-2</sup>s<sup>-1</sup> ( $1.8 \cdot 10^{23}$  m<sup>-2</sup>s<sup>-1</sup> for W). The maximum value of surface temperature in the central zone of plasma beam determined by IR camera was  $380-390^{\circ}$ C for W). The exposure time of deuterium plasma was 2000 s with resulting fluence of  $3.6-5 \cdot 10^{26}$  m<sup>-2</sup>.

### 2.3. In-situ LIBS and NRA measurements

For LIBS and NRA measurements, the samples were transferred from the linear plasma device into the TEAC without breaking the vacuum. This allowed to minimize H<sub>2</sub>O adsorption. First series of LIBS measurements were made in vacuum approximately one hour after deuterium loading in the time interval of 15-20 minutes (first for compact W-O, then for porous W-O, finally for thick W coating). LIBS measurements were followed by NRA measurements 2-4 hours after loading (Fig. 2). Most of the LIBS measurements were performed in the following day in two series (Fig. 2) These LIBS measurements were made at the background pressure of 1 mbar argon because the LIBS signal obtained in vacuum was very weak. Final series of NRA measurements were repeated almost 2 days after loading to check for possible changes in deuterium concentrations (Fig. 2).

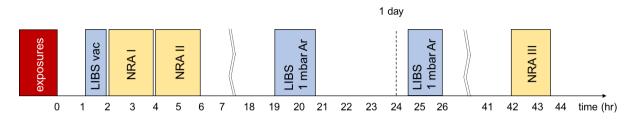


Figure 2. Time diagram showing different LIBS and NRA measurement series.

The description of the LIBS set-up can be found in earlier studies [20,33]. The used Nd:YAG laser operated at the fundamental wavelength of 1064 nm with pulse duration of 8ns. The laser beam was guided to the TEAC by mirrors of a 30 m long beamline and focused to the target by a lens with focal length of 1 m. The laser spot size corresponded to  $\approx 1.65 \text{ mm}^2$ . The fluence of the laser pulse at the target surface was kept at 20 J/cm<sup>2</sup> similarly to previous studies. The light emitted by laser plume was collected and imaged with a magnification of 1 onto a 0.8 mm diameter fiber by a 30 cm achromatic lens. A green alignment laser was used to align the recording fiber with a desired spot on the surface. The 30 m long fiber directed the light to a Czerny-Turner spectrometer with 1 m focal length and 1200 lines/mm grating coupled with an Andor iStar ICCD camera (DG340T-18F-03). A 20 nm wavelength range was preselected for detection of D and H emission at 656 nm. The delay of the detection window, relative to the LIBS laser pulse was 1300 ns and a gate width of 2000 ns was used for detection of the emitted plasma light.

D concentration profiles were determined by NRA analysis based on the nuclear reaction  $D(^{3}\text{He},p)^{4}\text{He}$ . These measurements were performed after LIBS measurements without venting in the same TEAC chamber. Two <sup>3</sup>He beam energies were used: 1000 keV to obtain a detailed profile of the surface (about 50 nm depth resolution) and 2500 keV to probe about 3 microns deep in tungsten. The dimensions of the beam spot were 2.5x5.5 mm<sup>2</sup>. The NRA detector was placed at a reaction angle of 172° with a solid angle of 7.9 msr. A 12.5 µm thick Mylar stopper foil was used to prevent that scattered <sup>3</sup>He particles could reach the detector.

# 3. Results and Discussion

The photos of samples exposed to deuterium flux for 2000 s are shown in figure 3. The order of the craters resulting from the LIBS measurements is marked on the figure by yellow numbers. NRA measurements were made at 5 different points on a vertical line through the center of samples (marked with blue rings). For compact W-O coating, there were clearly detectable darker rings on the coating surface while for porous W-O there was a slightly lighter region on the surface. The presence of these rings can be attributed to the location of central zone of plasma beam where the particle flux and temperature was highest. As the plasma beam did not hit the geometrical center of samples, the faint contrast of the plasma treated region complicates the comparison between results obtained by NRA and LIBS. The pure W coating had small specks attributable to flaking in most of the surface with the exception of the central region corresponding to the central part of plasma beam [28].

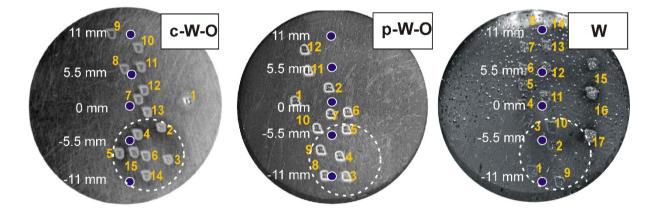


Figure 3. Photograph of the surface of W-O and W coatings exposed to deuterium flux for 2000 s. The blue circles note the approximate spots for NRA measurements while yellow numbers denote the positions of laser produced crates. The region on the surface exposed to central zone of the plasma beam is depicted by a dashed ring.

# 3.1. D and H intensities and elemental depth profiles by in-situ LIBS

Figure 4 shows the LIBS spectra at central wavelength of 656 nm collected from the regions of c-W-O and p-W-O coatings which were outside from the central zone of plasma beam (spot 11 for both coatings). The spectra obtained from W coating were comparable with the spectra obtained from c-W-O coating. The partly overlapping peaks at 656.1 nm and 656.28 nm, corresponding to  $D_{\alpha}$  and  $H_{\alpha}$  lines, were clearly detectable during the first 2-3 shots. The intensities of these lines,  $I_D$  and  $I_H$ , were obtained by fitting the measured spectra with

Lorentzian contours caused by Stark broadening (FWHM 0.07 to 0.1 nm). The intensity of H line,  $I_H$ , had comparable value in all coatings while  $I_D$  was clearly higher than  $I_H$  in the case of c-W-O and W coatings and smaller than  $I_H$  in the case of porous W-O coating.

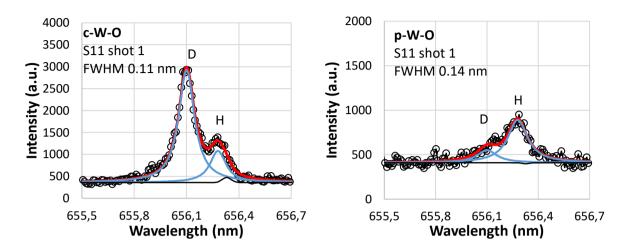
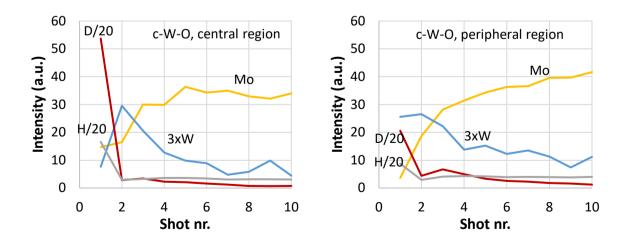


Figure 4. The in-situ LIBS spectrum of shot 1 recorded from spot 11 of c-W-O and p-W-O coating. These spots were corresponding to locations outside the central region of plasma beam.

Examples of depth profiles of H, D, W, Mo line intensities corresponding to central and peripheral zones of plasma beam are shown in figure 5. Intensities of W and Mo lines,  $I_W$  and  $I_{Mo}$ , were obtained by fitting W 657.4 nm and Mo 661.9 nm lines by Gaussians with the FWHM value corresponding to the apparatus function of spetrometer (0.05 nm).



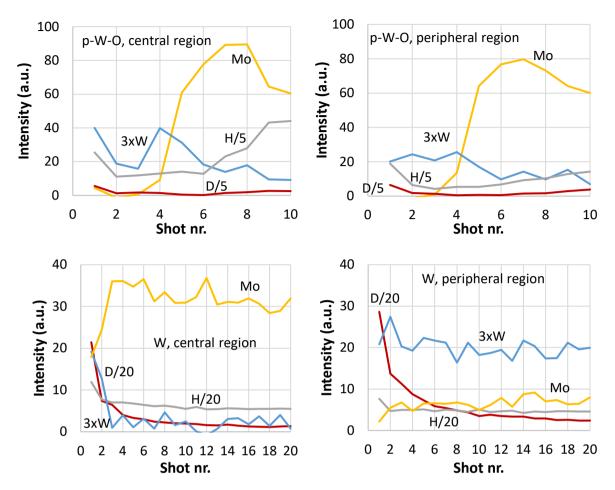


Figure 5. LIBS depth profiles of D, H, W and Mo line intensities averaged over spots in central or peripheral regions of plasma beam. The D and H intensities were divided by 5 or 20 and W intensity was multiplied by 3 to use the same scale for all lines.

The main observations for in-situ LIBS on c-W-O and p-W-O coatings are the following:

- $I_W$  was clearly detectable on the signal of background during 3-4 shots for c-W-O coating and 4-5 shots for p-W-O coating. The decrease of  $I_W$  was accompanied with the increase of  $I_{Mo}$ , indicating that the Mo substrate was encountered.
- Both *I*<sub>D</sub> and *I*<sub>H</sub> were highest during the first shot. In the case of c-W-O coating, the *I*<sub>D</sub> of first shot was higher in the region of coating exposed to the central zone of plasma beam while in case of p-W-O coating there was no clear correlation with the different regions on the coating.
- During subsequent shots, corresponding to increasing depth in the material, the *I*<sub>D</sub> remained above the noise level for 3-4 shots in the case of c-W-O coating while it reduced to the noise level for p-W-O coating. *I*<sub>H</sub> remained practically constant for c-W-O coating and somewhat increased for p-W-O coating.

 The ablation rates were determined by dividing the coating thickness with the number of laser shots required to reach the Mo substrate (3 shots for c-W-O coating and 5 shots for p-W-O coating). The resulting ablation rate was 460 nm/pulse for c-W-O coating and 320 nm/pulse for p-W-O coating.

The observed results obtained for c-W-O and p-W-O coatings are similar with our earlier exsitu LIBS study with similar coatings [28]. Main difference from our earlier ex-situ study is in H intensity which was previously considerably higher and complicated the determination of D inside of the coating.

The elemental depth profiles of the 10  $\mu$ m thick W coating are described separately from W-O coatings because they were markedly different:

- *I<sub>D</sub>* decreased approximately monotonically with increasing shot number and reached a stable value after about 15 shots. *I<sub>W</sub>* was somewhat higher during the first shot and then remained nearly constant.
- The  $I_W$  and  $I_{Mo}$  obtained from the W coating depended on the region of the coating. In the region subjected to the central zone of the plasma beam,  $I_{Mo}$  was higher than  $I_W$  even though the number of applied shots (20) was too small to reach the Mo substrate. In the peripheral region the  $I_W$  dominated and remained nearly constant during the applied 20 shots.

The origin of Mo in the W coating is unclear and requires additional studies. One possible explanation is the erosion of Mo substrate exposed by the flaking of W coating as suggested by figure 3. The eroded Mo is either implanted or deposited in the central zone of the plasma beam. An additional series of measurements was made with a bulk W sample, which was also exposed to deuterium plasma at similar conditions. In this case, the Mo signal was missing from LIBS spectra and the intensity of W lines remained same along the surface and through the depth profile. This result supports the proposed origin of Mo from the substrate of the coatings.

# 3.2. Depth profiles of normalized LIBS D intensities

For subsequent comparison of LIBS D depth profiles with NRA D concentrations,  $I_D$  was normalized either by  $I_H$  or averaged  $I_{Mo}$  value in the substrate. Normalizing was used to account for possible differences in alignment of the LIBS spectral registration system relative to the laser induced plasma plume for different spots and differences in laser plasma plume properties caused by different coating materials. The latter effect can be taken into account by using spectral lines of elements originating from the coating.  $I_W$  would be most preferable for normalizing because W content in the coating is known but unfortunately  $I_W$  was too weak and noisy. Therefore,  $I_H$  was used for normalizing of  $I_D$  even though it had several drawbacks: the concentration of hydrogen in the coating is not known, it varied in different coating regions and it had a higher intensity on the coating surface (Fig. 5). As an alternative to  $I_H$ , averaged  $I_{Mo}$  value originating from the substrate was used for normalizing because this line remained similar during different measurement series and did not depend on the region of the coating. However, the Mo signal from the substrate could only be determined for thinner W-O coatings.

Depth profiles based on the  $I_D/I_H$  ratio are shown in upper panel of figure 6. The ratios obtained from the coating regions exposed to the central zone of plasma beam were considerably lower when compared to the ratios correpsonding to peripheral zone. Highest  $I_D/I_H$  ratios were obtained in pure W coating followed by c-W-O coating. The  $I_D/I_H$  ratios of p-W-O coatings were the lowest and practically at the noise level. It should be noted that the  $I_H$  value of first shot was higher than the values of subsequent shots and therefore the  $I_D/I_H$ ratio of first shot is expectedly underestimated.  $I_D/I_{Mo}$  ratios shown in the lower panel of figure 6 for W-O coatings used  $I_{Mo}$  value averaged over shots 6 to 10 for normalizing.  $I_D/I_{Mo}$ ratios are mostly consistent with  $I_D/I_H$  ratios showing higher D content in coating regions corresponding to peripheral zone of plasma beam and an order of magnitude smaller D content in p-W-O coatings.

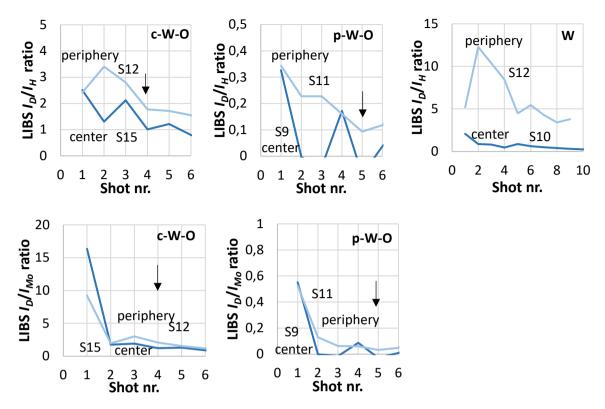


Figure 6. Depth profiles of  $I_D/I_H$  and  $I_D/I_{Mo}$  ratios obtained from different regions of plasma exposed coatings. Dark blue corresponds to the region exposed to the center of plasma beam while light blue corresponds to peripheral region. The arrows show the expected coating/substrate interface for W-O coatings.

#### 3.3. Deuterium depth profiles by in-situ NRA

NRA deuterium depth profiles measured 2-4 hours after the end of a D plasma exposure are shown on figure 7.

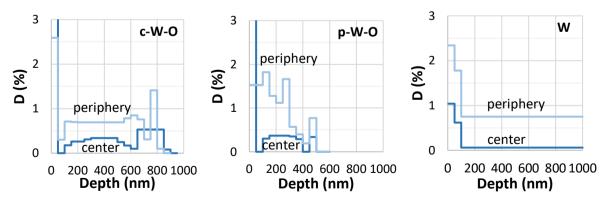


Figure 7. NRA deuterium depth profiles in c-W-O, p-W-O and W coatings obtained at central and peripheral zones of plasma exposed coatings. Dark blue corresponds to the region exposed to the center of plasma beam (5.5 mm) while light blue corresponds to peripheral

region (-5.5 mm). The y-axis is limited to 3 % to present more clearly D retention deeper in the coating.

Main NRA results are described below together with comparison with LIBS results:

- For all coatings, the deuterium concentration was highest in the first 50-100 nm of the material and this surface concentration depended on the composition of the coating and the measurement location on the coating. Deuterium concentration in this thin surface layer varied between 2-10 at. % in the c-W-O coating, 2-4 % in the p-W-O coating and 1-3 % in pure W coating. Higher D surface concentration observed by NRA is consistent with the higher D intensity observed during the first LIBS shot. Highest D concentration in the surface layer of 50-100 nm was also observed in our earlier SIMS study carried out with similar c-W-O and p-W-O coatings exposed in Magnum-PSI [28].
- In the case of c-W-O coating, the highest surface D concentration was observed in the region exposed to the central zone of plasma beam while for other coatings there was no clear correlation between surface concentration and the position on the coating. This tendency was also observed by LIBS.
- Deeper in the coating, the deuterium concentration remained below 2 % for all coatings and depended mostly on the coating region. Similar to the LIBS results, lower deuterium concentrations were determined in the regions which were exposed to the central zone of plasma beam.
- The main discrepancy between NRA and LIBS results was in the relative values of the average D content in different coatings. NRA showed highest D concentration in p-W-O coating while according to LIBS, this coating had an order of magnitude lower I<sub>D</sub>/I<sub>H</sub> and I<sub>D</sub>/I<sub>Mo</sub> ratio when compared to other coatings. The differences between LIBS and NRA results were almost comparable for c-W-O and W coatings.
- The retention depth of deuterium was approximately 900 nm for c-W-O and 500 nm for p-W-O coating. The D retention depth for pure W coating was at least 1  $\mu$ m, which was the depth limit for NRA. In the case of c-W-O and W coating the LIBS depth profiles of  $I_D/I_{Mo}$  ratios comply with these D retention depths. In the case of p-W-O coating the  $I_D/I_{Mo}$  ratio was above the noise level only for the first shot while the ablation rate was approximately 300 nm/shot. One possible reason for this discrepancy is that the actual ablation rate of porous W-O layer of the coating is higher than the estimated ablation

rate of full coating which includes the dense W interlayer. Another reason may be the temperature increase in the the non-ablated part of the coating during the first laser shot which results in the D desorption facilitated by the porous material.

Comparison of LIBS and NRA results allowed to estimate the sensitivity limits of LIBS in the used LIBS system. This limit depended on the coating material. In the case of compact W-O coating and pure W coating, the limit was below 0.5 % which may be sufficient when compared to up to 5 % D found in JET [34].

# 3.4. D spatial distribution and retention by NRA

Spatial distribution of D concentration (in percentage) averaged over the first 200 up to 500 nm depth is shown in upper panel of figure 8. The results are shown for three NRA measurement series (Fig. 2), two series made in 2-6 hours interval after loading and third series 40-42 hours after loading. The results show that the outgassing of deuterium was not important in the used time-window as the deuterium retention remained the same in the limits of accuracy. This is in line with another study [21], where it was observed that most of the deuterium outgassing takes place during the first few hours after the loading.

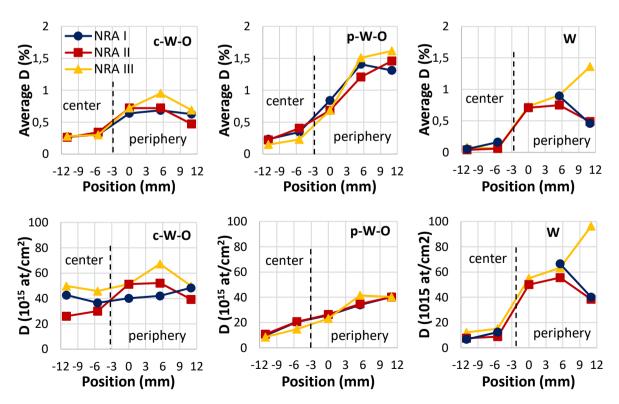


Figure 8. Spatial profiles of deuterium concentration integrated over 200 up to 500 nm depth (upper panel) and total retention of deuterium (lower panel) in c-W-O, p-W-O and W

coatings. NRA I and II series were made 2-6 hours after deuterium loading while III series was made 42 hours later (Fig. 2).

The average deuterium content in the coatings was lower in the region exposed to the center of the plasma beam. The difference between these regions was most notable for pure W and p-W-O coating. One reason for decreased D percentage in the coating region exposed to the central zone of plasma beam may be related to higher surface temperature in the central zone. Highest D retention is usually observed in the temperature range of 200-300°C while the retention decreases at higher temperatures [35]. In present study, the maximum temperature in the central zone was 380°C. The temperature decreases towards the outer regions of the Gaussian plasma beam [36] and 200-300°C is obtained outside of the central region of the coatings.

The total deuterium retention in 1  $\mu$ m thick layer of coating is shown in the lower panel of figure 8. D retention in the peripheral coating regions had comparable value (40-60·10<sup>15</sup> D at/cm<sup>2</sup>) for all coatings regardless of the composition. This suggests that the presence of oxygen has only small effect on the D retention at the conditions present in the peripheral zone of plasma beam. In the region corresponding to the central zone of plasma beam, the D retention remained higher for c-W-O coating. This high retention can be partially explained by the increased D content in the uppermost 50 nm thick surface layer of c-W-O coating. Our previous study with similar c-W-O coating showed the formation of a 50-100 nm thick surface layer with nanosized voids in the central zone of plasma exposure [28]. It has been suggested that such layer may retain higher amount of D while decreasing the retention deeper in the coating [37].

## 4. Conclusions

Present study carried out the in-situ LIBS measurements of D retention in porous W-O, compact W-O and pure W coatings loaded with D in Magnum-PSI linear plasma device. The obtained LIBS results were compared with in-situ NRA method.

According to NRA results, comparable deuterium retention was observed in both W-O coatings as well as in pure W coating. Except for 50-100 nm surface layer, higher D concentrations were determined in the regions of the coatings which were exposed to peripheral zone of plasma beam resulting in lower surface temperature. The differences in retention were less pronounced for different regions of c-W-O coating because the surface

layer had an order of magnitude higher D concentration in the region exposed to highest plasma flux in the central zone of plasma beam.

The results obtained by LIBS were partially in correlation with NRA results. LIBS  $I_D/I_{Mo}$  depth profiles were similar with NRA D concentration depth profiles and also reflected the spatial distribution of D along the sample surface.

Clear differences between the results of LIBS and NRA methods were observed for coatings with different composition. According to LIBS, porous W-O coating had much smaller  $I_D/I_H$ and  $I_D/I_{Mo}$  ratios when compared with other coatings while according to NRA the differences in D concentrations of different coatings were considerably less pronounced with p-W-O coating having highest D concentration. The difference is expectedly caused by different laser plasma plume properties for these coatings. These effects cannot be taken into consideration by normalizing  $I_D$  with  $I_{Mo}$  which originates from the substrate or with  $I_H$  which may vary for different coatings. In the future studies, the normalizing should be made by using intensities of elements which have known composition in the coating and give sufficiently strong intensity.

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### References

- G. Federici, P. Andrew, P. Barabaschi, J. Brooks, R. Doerner, A. Geier, A. Herrmann,
  G. Janeschitz, K. Krieger, A. Kukushkin, A. Loarte, R. Neu, G. Saibene, M. Shimada,
  G. Strohmayer, M. Sugihara, Key ITER plasma edge and plasma-material interaction
  issues, J. Nucl. Mater. 313–316 (2003) 11–22. doi:10.1016/S0022-3115(02)01327-2.
- [2] V. Philipps, J. Roth, A. Loarte, Plasma Physics and Controlled Fusion Key issues in plasma – wall interactions for ITER : a European approach Key issues in plasma – wall interactions for ITER : a, Plasma Phys. Control. Fusion. 45 (2003) A17–A30.
- [3] J. Roth, E. Tsitrone, A. Loarte, T. Loarer, G. Counsell, R. Neu, V. Philipps, S.Brezinsek, M. Lehnen, P. Coad, C. Grisolia, K. Schmid, K. Krieger, A. Kallenbach, B.

Lipschultz, R. Doerner, R. Causey, V. Alimov, W. Shu, O. Ogorodnikova, A. Kirschner, G. Federici, A. Kukushkin, Recent analysis of key plasma wall interactions issues for ITER, J. Nucl. Mater. 390–391 (2009) 1–9. doi:10.1016/j.jnucmat.2009.01.037.

- H.J.N. Van Eck, A.W. Kleyn, A. Lof, H.J. Van Der Meiden, G.J. Van Rooij, J.
  Scholten, P.A. Zeijlmans Van Emmichoven, Divertor conditions relevant for fusion reactors achieved with linear plasma generator, Appl. Phys. Lett. 101 (2012). doi:10.1063/1.4768302.
- [5] M.J. van de Pol, S. Alonso van der Westen, D.U.B. Aussems, M.A. van den Berg, S. Brons, H.J.N. van Eck, G.G. van Eden, H.J.W. Genuit, H.J. van der Meiden, T.W. Morgan, J. Scholten, J.W.M. Vernimmen, E.G.P. Vos, M.R. de Baar, Operational characteristics of the superconducting high flux plasma generator Magnum-PSI, Fusion Eng. Des. 136 (2018) 597–601. doi:10.1016/j.fusengdes.2018.03.033.
- [6] M. Mayer, J. Likonen, J.P. Coad, H. Maier, M. Balden, S. Lindig, E. Vainonen-Ahlgren, V. Philipps, Tungsten erosion in the outer divertor of JET, J. Nucl. Mater. 363–365 (2007) 101–106. doi:10.1016/j.jnucmat.2007.01.010.
- J. Likonen, K. Heinola, A. De Backer, S. Koivuranta, A. Hakola, C.F. Ayres, A. Baron-Wiechec, P. Coad, G.F. Matthews, M. Mayer, A. Widdowson, Deuterium trapping and release in JET ITER-like wall divertor tiles, Phys. Scr. 2016 (2016). doi:10.1088/0031-8949/T167/1/014074.
- [8] S. Masuzaki, M. Tokitani, T. Otsuka, Y. Oya, Y. Hatano, M. Miyamoto, R. Sakamoto, N. Ashikawa, S. Sakurada, Y. Uemura, K. Azuma, K. Yumizuru, M. Oyaizu, T. Suzuki, H. Kurotaki, D. Hamaguchi, K. Isobe, N. Asakura, A. Widdowson, K. Heinola, S. Jachmich, M. Rubel, Analyses of microstructure, composition and retention of hydrogen isotopes in divertor tiles of JET with the ITER-like wall, Phys. Scr. 2017 (2017). doi:10.1088/1402-4896/aa8bcc.
- [9] M. Balden, A. Manhard, S. Elgeti, Deuterium retention and morphological modifications of the surface in five grades of tungsten after deuterium plasma exposure, J. Nucl. Mater. 452 (2014) 248–256. doi:10.1016/j.jnucmat.2014.05.018.
- [10] O. V. Ogorodnikova, Effect of nanostructure on radiation tolerance and deuterium retention in tungsten, J. Appl. Phys. 122 (2017). doi:10.1063/1.4996096.
- [11] O.V. Ogorodnikova, C. Ruset, D. Dellasega, A. Pezzoli, M. Passoni, K. Sugiyama, Y. Gasparayan, V. Efimov, Deuterium retention in dense and disordered nanostructured tungsten coatings, J. Nucl. Mater. 507 (2018) 226–240.

- [12] V.K. Alimov, B. Tyburska, M. Balden, S. Lindig, J. Roth, K. Isobe, T. Yamanishi, Surface morphology and deuterium retention in tungsten oxide layers exposed to lowenergy, high flux D plasma, J. Nucl. Mater. 409 (2011) 27–32.
- [13] A. Pezzoli, D. Dellasega, V. Russo, A. Gallo, P.A.. Zeijlmans van Emmichoven, M. Passoni, E. Besozzi, D. Dellasega, V. Russo, C. Conti, M. Passoni, M.G. Beghi, A. Maffini, A. Uccello, D. Dellasega, M. Passoni, Thermal annealing and exposure to divertor-like deuterium plasma of tailored tungsten oxide coatings, J. Nucl. Mater. 463 (2015) 1041–1044. doi:10.1016/j.matdes.2018.107565.
- [14] A. Semerok, D. L'Hermite, J.-M. Weulersse, J.-L. Lacour, G. Cheymol, M. Kempenaars, N. Bekris, C. Grisolia, Laser induced breakdown spectroscopy application in joint European torus, Spectrochim. Acta Part B. 123 (2016) 121–128.
- [15] D. Zhao, C. Li, Z. Hu, C. Feng, Q. Xiao, R. Hai, P. Liu, L. Sun, D. Wu, C. Fu, J. Liu, N. Farid, F. Ding, G.-N. Luo, L. Wang, H. Ding, Remote in situ laser-induced breakdown spectroscopic approach for diagnostics of the plasma facing components on experimental advanced superconducting tokamak, Rev. Sci. Instrum. 89 (2018) 073501.
- [16] G. Maddaluno, S. Almaviva, L. Caneve, F. Colao, V. Lazic, L. Laguardia, P. Gasior,
  M. Kubkowska, Detection by LIBS of the deuterium retained in the FTU toroidal
  limiter, Nucl. Mater. Energy. 18 (2019) 208–211. doi:10.1016/j.nme.2018.12.029.
- [17] S. Almaviva, L. Caneve, F. Colao, V. Lazic, G. Maddaluno, P. Mosetti, A. Palucci, A. Reale, P. Gasior, W. Gromelski, M. Kubkowska, LIBS measurements inside the FTU vessel mock-up by using a robotic arm, Fusion Eng. Des. 157 (2020) 111685. doi:10.1016/j.fusengdes.2020.111685.
- [18] Q. Xiao, A. Huber, G. Sergienko, B. Schweer, P. Mertens, A. Kubina, V. Philipps, H. Ding, Application of laser-induced breakdown spectroscopy for characterization of material deposits and tritium retention in fusion devices, Fusion Eng. Des. 88 (2013) 1813–1817. doi:10.1016/j.fusengdes.2013.05.083.
- [19] C. Li, X. Wu, C. Zhang, H. Ding, G. De Temmerman, H.J. Van Der Meiden, Study of deuterium retention on lithiated tungsten exposed to high-flux deuterium plasma using laser-induced breakdown spectroscopy, Fusion Eng. Des. 89 (2014) 949–954. doi:10.1016/j.fusengdes.2014.04.071.
- [20] K. Piip, G. De Temmerman, H.J. Van Der Meiden, A. Lissovski, J. Karhunen, M. Aints, A. Hakola, P. Paris, M. Laan, J. Likonen, I. Jõgi, J. Kozlova, H. Mändar, LIBS analysis of tungsten coatings exposed to Magnum PSI ELM-like plasma, J. Nucl.

Mater. 463 (2015). doi:10.1016/j.jnucmat.2014.11.017.

- [21] X. Jiang, G. Sergienko, B. Schweer, S. Möller, M. Freisinger, A. Kreter, S. Brezinsek, C. Linsmeier, An upgraded LIBS system on linear plasma device PSI-2 for in situ diagnostics of plasma-facing materials, Fusion Eng. Des. 146 (2019) 96–99. doi:10.1016/j.fusengdes.2018.11.044.
- [22] Z. Hu, C. Li, Q. Xiao, P. Liu, F. Ding, H. Mao, J. Wu, D. Zhao, H. Ding, G.N. Luo, Preliminary results of in situ laser-induced breakdown spectroscopy for the first wall diagnostics on EAST, Plasma Sci. Technol. 19 (2017). doi:10.1088/2058-6272/19/2/025502.
- [23] P. Wang, W. Jacob, L. Gao, T. Dürbeck, T. Schwarz-Selinger, Comparing deuterium retention in tungsten films measured by temperature programmed desorption and nuclear reaction analysis, Nucl. Instruments Methods Phys. Res. Sect. B Beam Interact. with Mater. Atoms. 300 (2013) 54–61. doi:10.1016/j.nimb.2013.01.057.
- [24] M.H.J. 'T Hoen, M. Balden, A. Manhard, M. Mayer, S. Elgeti, A.W. Kleyn, P.A. Zeijlmans Van Emmichoven, Surface morphology and deuterium retention of tungsten after low- and high-flux deuterium plasma exposure, Nucl. Fusion. 54 (2014). doi:10.1088/0029-5515/54/8/083014.
- [25] M. Oya, H.T. Lee, Y. Ueda, H. Kurishita, M. Oyaidzu, T. Hayashi, N. Yoshida, T.W. Morgan, G. De Temmerman, Surface morphology changes and deuterium retention in Toughened, Fine-grained Recrystallized Tungsten under high-flux irradiation conditions, J. Nucl. Mater. 463 (2015) 1037–1040. doi:10.1016/j.jnucmat.2014.11.124.
- [26] M. Kelemen, A. Založnik, P. Vavpetič, M. Pečovnik, P. Pelicon, A. Hakola, A. Lahtinen, J. Karhunen, K. Piip, P. Paris, M. Laan, K. Krieger, M. Oberkofler, H. van der Meiden, S. Markelj, Micro-NRA and micro-3HIXE with 3He microbeam on samples exposed in ASDEX Upgrade and Pilot-PSI machines, Nucl. Instruments Methods Phys. Res. Sect. B Beam Interact. with Mater. Atoms. 404 (2017) 179–184. doi:10.1016/j.nimb.2017.01.072.
- [27] K. Schmid, J. Bauer, T. Schwarz-Selinger, S. Markelj, U. V. Toussaint, A. Manhard,
  W. Jacob, Recent progress in the understanding of H transport and trapping in W,
  Phys. Scr. 2017 (2017). doi:10.1088/1402-4896/aa8de0.
- [28] I. Jõgi, P. Paris, J. Kozlova, H. Mändar, M. Passoni, D. Dellasega, A. Hakola, H.J. van der Meiden, LIBS study of ITER relevant tungsten-oxygen coatings exposed to deuterium plasma in Magnum-PSI, J. Nucl. Mater. In Press (2020) 152660.
- [29] D. Dellasega, G. Merlo, C. Conti, C.E. Bottani, M. Passoni, Nanostructured and

amorphous-like tungsten films grown by pulsed laser deposition, J. Appl. Phys. 112 (2012) 084328. doi:10.1063/1.4761842.

- [30] R. Mateus, D. Dellasega, M. Passoni, Z. Siketić, I. Bogdanović Radović, A. Hakola, E. Alves, Helium load on W-O coatings grown by pulsed laser deposition, Surf. Coatings Technol. 355 (2018) 215–221. doi:10.1016/j.surfcoat.2018.02.089.
- [31] C. Costin, V. Anita, G. Popa, J. Scholten, G. De Temmerman, Tailoring the charged particle fluxes across the target surface of Magnum-PSI, Plasma Sources Sci. Technol. 25 (2016). doi:10.1088/0963-0252/25/2/025023.
- [32] H.J. Van Der Meiden, A.R. Lof, M.A. Van Den Berg, S. Brons, A.J.H. Donné, H.J.N. Van Eck, P.M.J. Koelman, W.R. Koppers, O.G. Kruijt, N.N. Naumenko, T. Oyevaar, P.R. Prins, J. Rapp, J. Scholten, D.C. Schram, P.H.M. Smeets, G. Van Der Star, S.N. Tugarinov, P.A.Z. Van Emmichoven, Advanced Thomson scattering system for high-flux linear plasma generator, Rev. Sci. Instrum. 83 (2012). doi:10.1063/1.4768527.
- [33] H.J. Van Der Meiden, M.A. Van Den Berg, S. Brons, H. Ding, H.J.N. Van Eck, M.H.J. 'T Hoen, J. Karhunen, T.M. De Kruif, M. Laan, C. Li, A. Lissovski, T.W. Morgan, P. Paris, K. Piip, M.J. Van De Pol, R. Scannell, J. Scholten, P.H.M. Smeets, C. Spork, P.A. Zeijlmans Van Emmichoven, R. Zoomers, G. De Temmerman, Laser-based diagnostics applications for plasma-surface interaction studies, J. Instrum. 8 (2013). doi:10.1088/1748-0221/8/11/C11011.
- [34] M. Mayer, S. Krat, A. Baron-Wiechec, Y. Gasparyan, K. Heinola, S. Koivuranta, J. Likonen, C. Ruset, G. De Saint-Aubin, A. Widdowson, Erosion and deposition in the JET divertor during the second ITER-like wall campaign, Phys. Scr. 2017 (2017). doi:10.1088/1402-4896/aa8ff9.
- [35] V.K. Alimov, W.M. Shu, J. Roth, K. Sugiyama, S. Lindig, M. Balden, K. Isobe, T. Yamanishi, Surface morphology and deuterium retention in tungsten exposed to low-energy, high flux pure and helium-seeded deuterium plasmas, Phys. Scr. T. T138 (2009). doi:10.1088/0031-8949/2009/T138/014048.
- [36] K. Piip, H.J. van der Meiden, L. Hämarik, J. Karhunen, A. Hakola, M. Laan, P. Paris, M. Aints, J. Likonen, K. Bystrov, J. Kozlova, A. Založnik, M. Kelemen, S. Markelj, LIBS detection of erosion/deposition and deuterium retention resulting from exposure to Pilot-PSI plasmas, J. Nucl. Mater. 489 (2017) 129–136. doi:10.1016/j.jnucmat.2017.03.044.
- [37] J. Bauer, T. Schwarz-Selinger, K. Schmid, M. Balden, A. Manhard, U. Von Toussaint, Influence of near-surface blisters on deuterium transport in tungsten, Nucl. Fusion. 57

(2017). doi:10.1088/1741-4326/aa7212.